

Synthesis and Structural Characterization of Cr₂O₃ Nanoparticles Prepared by Using Cr(NO₃)₃·9H₂O and Triethanolamine Under Microwave Irradiation

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Abstract

In this study, Chromium oxide nanoparticles (Cr₂O₃-NPs) was prepared by sol-gel method, using Cr(NO₃)₃·9H₂O as a Precursor material, triethanolamine (TEA) as a template and water as green solvent under microwave irradiation . X- ray diffraction (XRD), Scanning electron microscopy (SEM), Rietveld refinement and Fourier Transform Infrared spectroscopy (FTIR) were employed to characterize the average crystallite size, crystalline structure, morphology, unit cell parameters and chemical groups of the annealed powder. From XRD data and Dicvol 91 software analysis, the crystal system was found to be rhombohedral structure for samples. The average crystallite size is calculated by using Debye-Scherrer formula, crystallites size and strain are calculated by using Williamson-Hall formula. The effect of particle size on the crystal lattice distortion ratio, specific surface area, dislocation density and number of unit cell was discussed. A good agreement is between XRD data and FIR data. The qualitative phase analysis is performed by Rietveld analysis using “Fullprof” program of Cr₂O₃-NPs.

Keywords: Cr₂O₃, Nanoparticles, Sol – Gel , Structural Properties, Rietveld refinement.

1. Introduction

Nano size materials exhibit unique electronic, magnetic, optical, catalytic and medicinal properties as compared with the traditional and commercial bulk materials. It is due to its quantum size effect, large surface to volume ratio [1]. Chromium (III) oxide belongs to rhombohedral crystal system and is of great interest due to its easiness of preparation and it has a wide range of applications such as heterogeneous catalyst [1], pigments to reflect infrared radiation [2], solar energy application [3], high temperature resistant materials because of its high melting temperature ($\approx 2300^\circ\text{C}$) [3,4], liquid crystal displays [5], wear resistance [6] and coating material [7]. Different methods are available to prepare Cr₂O₃-NPs such as sol-gel technique [8], sonochemical method [9], precipitation-gelation [10], mechanochemical process [11], microwave plasma [12] and gas condensation [13], etc.

In the present study, we have synthesized (Cr₂O₃-NPs) under microwave irradiation using TEA as template and water as a green solvent.

2- Experimental Part

2-1 Synthesis

All chemicals and reagents were of synthetic grade and used without further purification. In a typical procedure, 50 ml of Cr(NO₃)₃·9H₂O (0.2M) (99.5%, Fluka) aqueous solution was mixed with appropriate amount of triethanolamine (C₆H₁₅NO₃) (99.5%, Fluka) as template (20 and 30 mmol). After stirring for 1 hour, the mixture was placed under microwave irradiation for 3 min. The green solid product was centrifuged and dried in air at room temperature . Six samples were prepared , first three by adding 20 mmol of (TEA) and second there by adding 30 mmol (TEA) , then annealed at 500 °C and 600°C for 2 h ,the details of these samples are shown in table (1).

Table 1. Concentration of triethanolamine (TEA) at various annealing temperature of Cr₂O₃-NPs

| (TEA) | Annealing temperature | Sample |
|---------|-----------------------|--------|
| 20 mmol | As prepared | S1 |
| | Annealed at 500°C | S2 |
| | Annealed at 600°C | S3 |
| 30 mmol | As prepared | S4 |
| | Annealed at 500°C | S5 |
| | Annealed at 600°C | S6 |

2-2 Characterization

The crystalline structure of the prepared powders was determined by X-ray diffraction (Shimadzu XRD-6000) with Cu $K\alpha_1$ radiation. The average crystallite size (D) of each powder was estimated from the Scherrer formula and Williamson-Hall formula.

The powder morphology was observed using a (VEGA\Easy Probe) scanning electron microscope. Fourier transformation infrared spectroscopy analysis of annealed powders was carried out in Shimadzu equipment (IRAffinity-1 FTIR Spectrophotometer) for studying the chemical groups of the annealed powder. Rietveld analysis was carried out to calculate unit cell parameters.

3- Results and Discussion

3-1 XRD measurement

The XRD pattern of all the samples, Cr_2O_3 nanoparticles (NPs) as prepared, annealed at 500°C and 600°C by using (20 and 30) mmol of triethanolamine (TEA) are shown in figure (1). They were compared with JCPDS file No.38-1479 and no crystalline impurities were detected. The observed diffraction peaks correspond to the rhombohedral crystalline Cr_2O_3 (Eskolaite) with space group ($R\bar{3}c$). The results of diffraction angle (2θ), full width at half maximum (FWHM) β , spacing between crystal planes (d) for three strongest peaks are shown in table (2).

3-2 Crystallite Size Calculation

The average crystallite size for synthesized Cr_2O_3 was determined by Debye-Scherrer's equation [14-17]:

$$D = K\lambda / \beta_D \cos\Theta \quad \dots\dots(1)$$

where D: crystallite size, K: constant, λ : wavelength of Cu $k\alpha$ radiation and β_D : is the broadening due to small crystallite size.

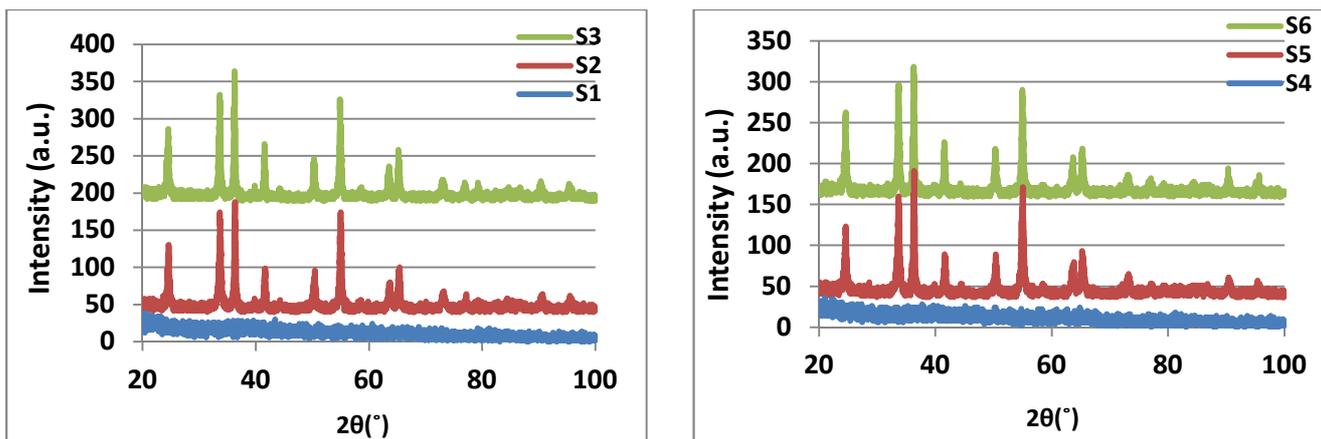


Figure 1. XRD patterns of Cr_2O_3 -NPs.

The crystallite size and lattice strain induced in powders arising from defects like dislocation, twinning, imperfection and distortion was determined by Williamson-Hall equation [18]:

$$\beta_{hkl} \cos\Theta = (K\lambda / D) + (4\epsilon \sin\Theta) \quad \dots\dots(2)$$

W-H plot is shown in figure (2). It is plotted with $\sin\Theta$ on the x-axis and $\beta_{hkl} \cos\Theta$ on the y-axis (β_{hkl} in radian). Table (3) shows strain and crystallite size according to W-H and crystallite size according to Debye-Scherrer for Cr_2O_3 nanoparticles.

Table 2. XRD data of Cr₂O₃-NPs

| Sample | 2θ(°) | d(Å) | β(°) | I/I ₁ |
|--------|---------|---------|-------|------------------|
| S1 | 22.3674 | 3.97153 | 0.048 | 100 |
| | 29.5529 | 3.02021 | 0.078 | 100 |
| | 30.4793 | 2.93049 | 0.073 | 100 |
| S2 | 36.335 | 2.470 | 0.304 | 100 |
| | 54.978 | 1.668 | 0.395 | 91 |
| | 33.706 | 2.656 | 0.345 | 89 |
| S3 | 36.264 | 2.475 | 0.304 | 100 |
| | 54.901 | 671 | 0.310 | 95 |
| | 33.623 | 2.663 | 0.382 | 88 |
| S4 | 35.082 | 2.555 | 0.070 | 100 |
| | 35.697 | 2.513 | 0.120 | 100 |
| | 39.7390 | 2.26640 | 0.110 | 100 |
| S5 | 36.318 | 2.471 | 0.329 | 100 |
| | 54.971 | 1.669 | 0.424 | 87 |
| | 33.675 | 2.659 | 0.400 | 76 |
| S6 | 36.296 | 2.473 | 0.295 | 100 |
| | 54.942 | 1.670 | 0.313 | 92 |
| | 33.667 | 2.660 | 0.344 | 91 |

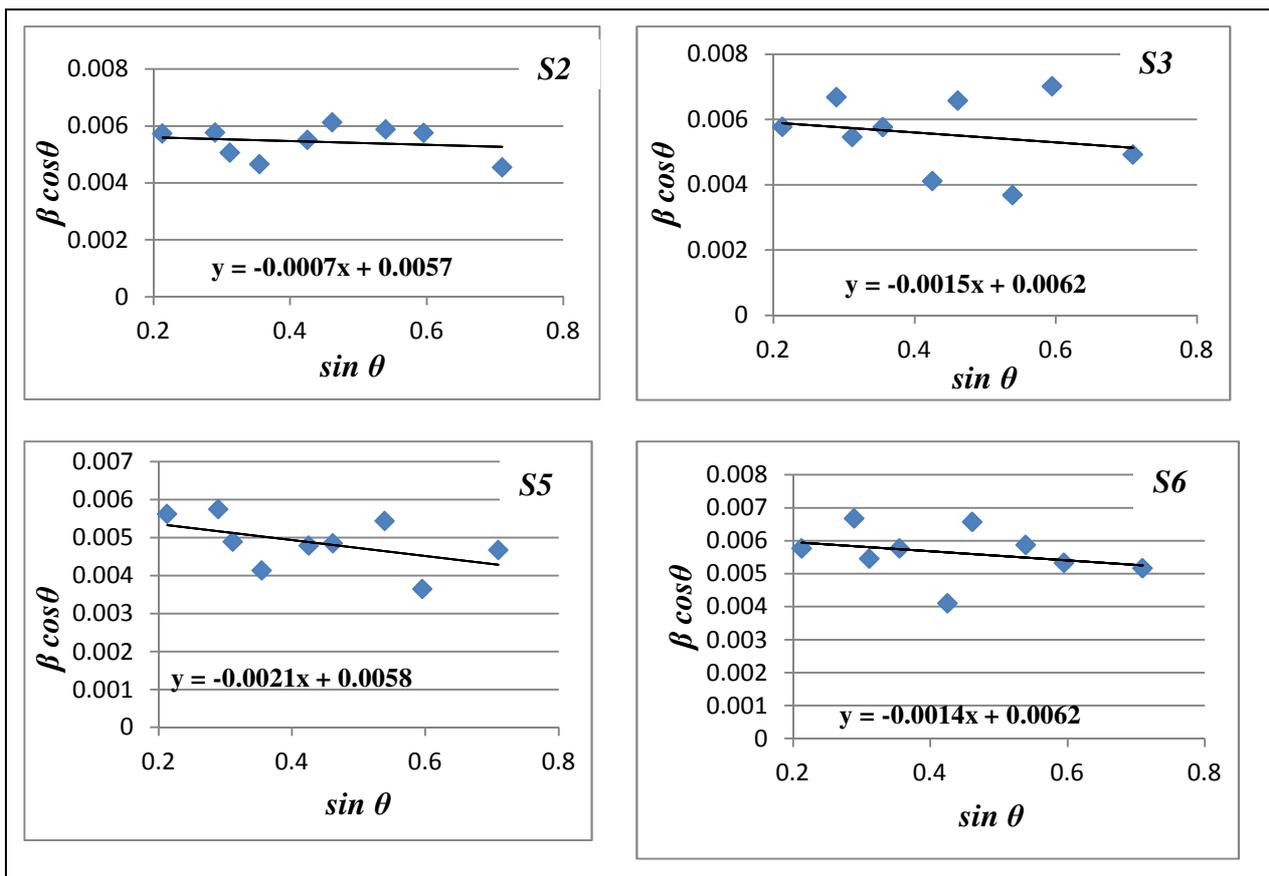


Figure 2. Williamson Hall plot of Cr₂O₃-NPs

Table 3. Strain and crystallite size according to W- H and crystallite size according to D-S of Cr₂O₃-NPs

| Sample | D (W-H)nm | ε (W-H)*10 ⁻⁴ | D (D-S) nm |
|--------|-----------|--------------------------|------------|
| S2 | 25.7 | -1.75 | 23.1 |
| S3 | 26.5 | -3.75 | 24.3 |
| S5 | 27.9 | -5.25 | 22.3 |
| S6 | 29.2 | -3.5 | 23.9 |

From table (3) it shows that the average crystallite size increases with the increase of annealing temperature and concentration of (TEA) because of decreases in the density of nucleation centers, thus a smaller number of centers starts to grow, resulting in large crystallites. Generally the average crystallite size depends on the annealing temperature[19,20]. The difference between crystallite size according to W-H and crystallite size according to Debye-Scherrer occurs because W-H method takes into account the lattice strain of crystallite arising from defects like dislocation, twinning, imperfection and distortion. A negative value for the strain is due to lattice shrinkage [21].

The crystal lattice distortion ratio of Cr₂O₃-NPs at different annealing temperatures and concentrations of (TEA) are calculated by using equation (3)[26]:

$$\beta_{hkl}^2 \cos^2\Theta = (4\lambda^2/\pi^2 D^2) + 32\langle\epsilon^2\rangle \sin^2\Theta \dots\dots(3)$$

From equations (1 and 3), it can be deduced the lattice distortion ratio used in the form of equation (4):

$$\sqrt{\langle\epsilon^2\rangle} = \frac{1}{D} \frac{1}{\sin\theta} \frac{\pi}{\lambda} \sqrt{\frac{\pi^2 k^2 - 4}{32}} \dots\dots(4)$$

Where $\langle\epsilon^2\rangle^{1/2}$ is the crystal lattice distortion ratio, it is shown in table (4) for S2, S3, S5 and S6 of Cr₂O₃-NPs.

Table 4. Crystal lattice distortion ratio of Cr₂O₃-NPs

| Sample | D (W-H) nm | lattice distortion ratio |
|--------|------------|--------------------------|
| S2 | 25.7 | 0.0268 |
| S3 | 26.5 | 0.0230 |
| S5 | 27.9 | 0.0251 |
| S6 | 29.2 | 0.237 |

The crystal lattice distortion ratio decreases with the increase of the average crystallite size and annealing temperature

3-3 Specific Surface Area (SSA)

SSA is the area per unit mass, it is a factor to determine bulk rates of such reactions (unit m²/g) [22]. It is very important in the nanoparticles because their large surface to volume ratio with a decrease in particle size[23]. SSA is used in materials to determine their types and properties and is also used in case of reactions on surfaces, heterogeneous catalysis and adsorption[24].

Mathematically, SSA can be determined by using equation (5)[25]. Specific surface area of Cr₂O₃ nanoparticles is shown in table (5).

$$SSA=6 * 10^3 / D_p \rho \dots\dots(5)$$

Where ρ is the density of Cr₂O₃ (5.22 g.cm⁻³).

From table (5); it is clear that SSA increases when the particle size and annealing temperature decreases because of the degree of agglomeration by the effect of annealing occurring in the material.

Table 5. Specific surface area of Cr₂O₃-NPs

| Sample | D (W-H) nm | SSA (m ² . g ⁻¹) | D (D-S) nm | SSA (m ² . g ⁻¹) |
|--------|------------|---|------------|---|
| S2 | 25.7 | 44.7 | 23.1 | 49.7 |
| S3 | 26.5 | 43.3 | 24.3 | 47.3 |
| S5 | 27.9 | 41.1 | 22.3 | 51.5 |
| S6 | 29.2 | 39.2 | 23.9 | 48.1 |

3-4 Rietveld Refinement

Rietveld refinement is a technique used to solve a structure crystalline materials from the powder diffraction data. The results powder diffraction from X-ray are characterized by reflections as a peak, intensity at certain position. The position, height and width for peak can be used to determine the material's structure. The Rietveld method uses a least squares to reduce the difference between the calculated and observed patterns, if the value of χ^2 is close to 1, that indicates the appropriate quality between observation and calculated data [27–29]. Rietveld refinements of Cr₂O₃-NPs for S2, S3, S5 and S6 are shown in figure (3) and the results of indexing and refinement are shown in table (6).

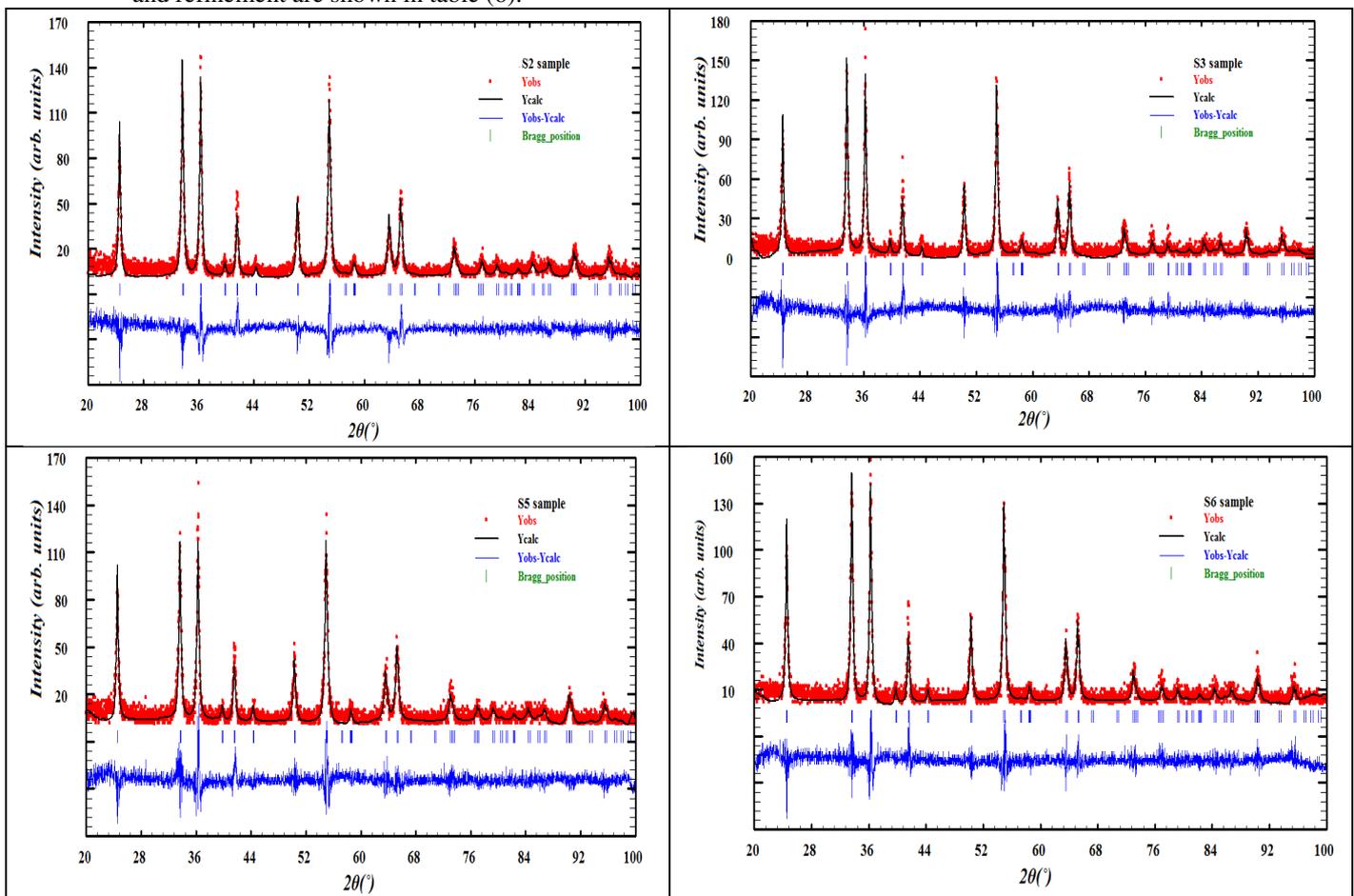


Figure 3. Rietveld refinements of Cr₂O₃-NPs for S2, S3, S5 and S6 samples.

The experimental points are plotted as dots (.) and theoretical data are shown as solid line. The difference between theoretical and experimental data is shown in the bottom line of each figure. The vertical lines represent the Bragg's allowed peaks.

Table 6. The results of indexing and refinement for Cr₂O₃-NPs.

| Sample | Unit cell parameters (Å) | | Crystal System | Space group | Angles (°) | | Atomic Position | | | | χ^2 |
|--------|--------------------------|--------|----------------|---------------|----------------|----------|-----------------|--------|-----|--------|----------|
| | a=b | c | | | $\alpha=\beta$ | γ | atom | x | y | z | |
| S2 | 4.948 | 13.582 | Rhom. | R $\bar{3}$ c | 90 | 120 | Cr | 0.0 | 0.0 | 0.3478 | 1.93 |
| | | | | | | | O | 0.3047 | 0.0 | 0.25 | |
| S3 | 4.951 | 13.593 | Rhom. | R $\bar{3}$ c | 90 | 120 | Cr | 0.0 | 0.0 | 0.3472 | 1.80 |
| | | | | | | | O | 0.3077 | 0.0 | 0.25 | |
| S5 | 4.949 | 13.580 | Rhom. | R $\bar{3}$ c | 90 | 120 | Cr | 0.0 | 0.0 | 0.3447 | 1.53 |
| | | | | | | | O | 0.3134 | 0.0 | 0.25 | |
| S6 | 4.952 | 13.589 | Rhom. | R $\bar{3}$ c | 90 | 120 | Cr | 0.0 | 0.0 | 0.3469 | 1.57 |
| | | | | | | | O | 0.3043 | 0.0 | 0.25 | |

From figure (3) it is noticed that peak width of decreases with the increase of the annealing temperature are due to growth of crystallite and construction to larger clusters.

3-5 Dislocation Density

Dislocation density (δ) is the length of the dislocations present per unit area (lines/m²). It is an important property for material. Many of the properties of materials are affected by dislocation, it is a crystallographic defect, or irregularity, within a crystal structure[30]. Dislocation density depends on the materials microstructure. The dislocation density is measured by direct methods, such as transmission electron microscopy (TEM) techniques, and by indirect methods, such as X-ray diffraction (XRD) or neutron techniques [31, 32]. The dislocation density (δ) in the samples is calculated by using equation (6)[9, 26].

$$\delta = \frac{15 \beta \cos \theta}{4 a D} \dots\dots(6)$$

Where, δ : dislocation density and a : lattice constant (nm).

And also can be calculated from equation (7).

$$\delta = \frac{1}{D^2} \dots\dots\dots(7)$$

The number of unit cell is calculated from equation (8).

$$n = \pi (4/3) \cdot (D/2)^3 \cdot (1/V) \dots\dots(8)$$

where n: the number of unit cell and V: is the cell volume.

The number of unit cell Vs dislocation density and versus particle size of Cr₂O₃-NPs for S2, S3, S5 and S6 are shown in figure (4 and 5) respectively.

From figures (4 and 5); it shows that dislocation density decreases and the number of unit cells increases with the increase in particle size and calcination temperatures that leads to increase the crystal growth and decrease the defects in crystallites, that means the crystals with larger dislocation density were harder [34].

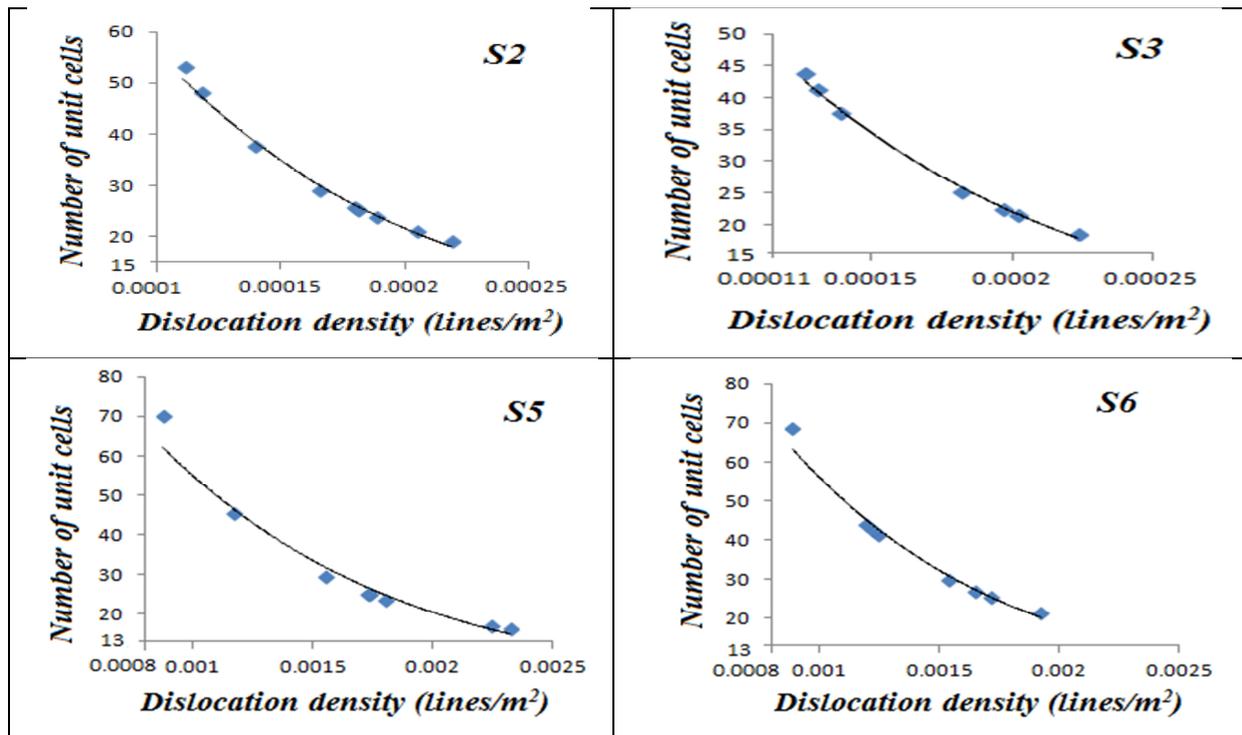


Figure 4. Number of unit cell versus dislocation density of Cr₂O₃-NPs

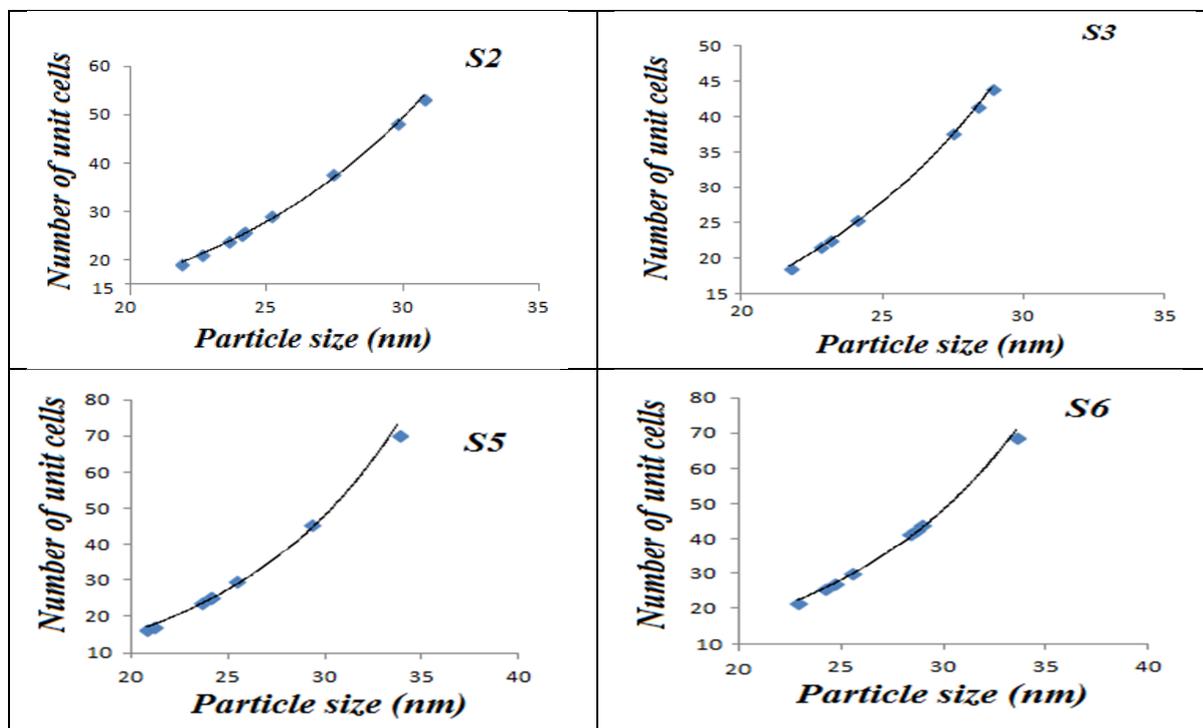


Figure 5. Number of unit cell versus particle size of Cr₂O₃-NPs

3-6 Fourier Transform Infrared Spectroscopy Analysis

Figure (6) shows the typical FTIR spectra of Cr_2O_3 -NPs . The spectra represents five sharp absorption bands at 424 cm^{-1} , 487 cm^{-1} , 547 cm^{-1} , 607 cm^{-1} and 638 cm^{-1} respectively. All the observed peaks are in well agreement with the reported results. In general, the metal oxide symbolizes the peaks below 1000 cm^{-1} as a result of inter-atomic vibrations. In the present investigation , the observed peaks illustrate the characteristic peaks of Cr–O bond stretching vibration . As the spectrum reveals the corresponding vibration bands of Cr–O only which in turn depicts the successful removal of impurity contents during calcinations and the high purity of the as-grown Cr_2O_3 -NPs. High intensity of the peaks of Cr_2O_3 bands indicates the good crystalline nature of the materials [3].

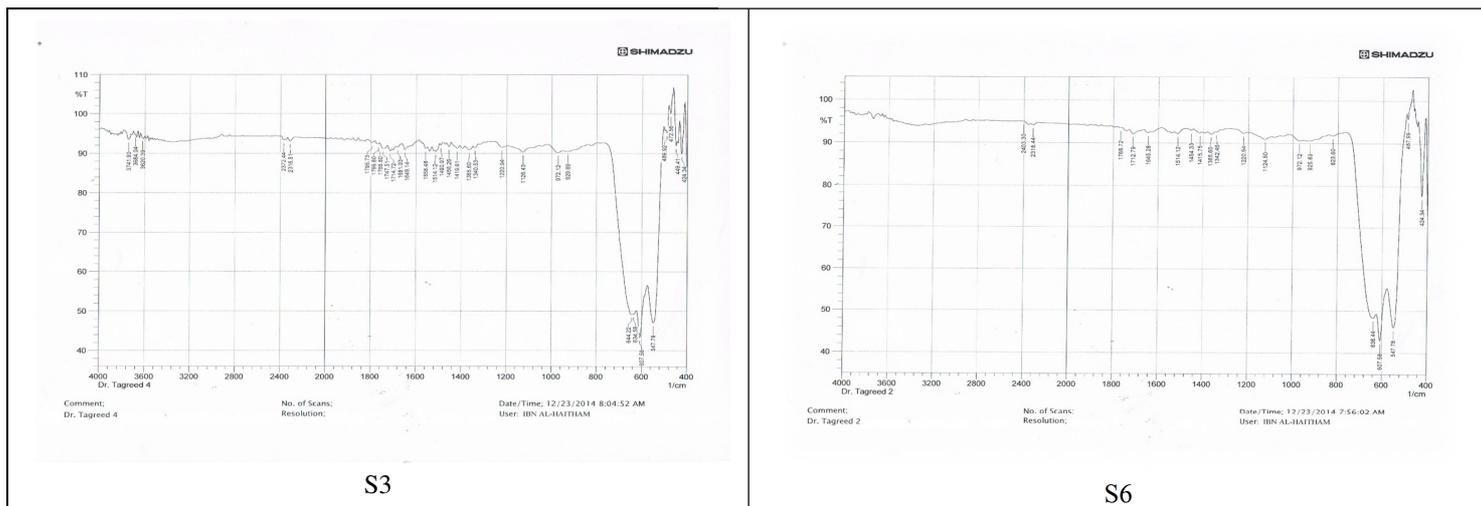


Figure 6. FTIR spectra of Cr_2O_3 -NPs

3-7 SEM images analysis

The SEM micrographs for Cr_2O_3 -NPs S3 and S6 are shown in figure (7), this Figure shows all particles exhibit a spherical shape with a high degree of agglomeration among fine particles. Figure (7b) shows the increase concentration of (TEA) , the average crystallite size increase and more agglomeration occurs among particles.

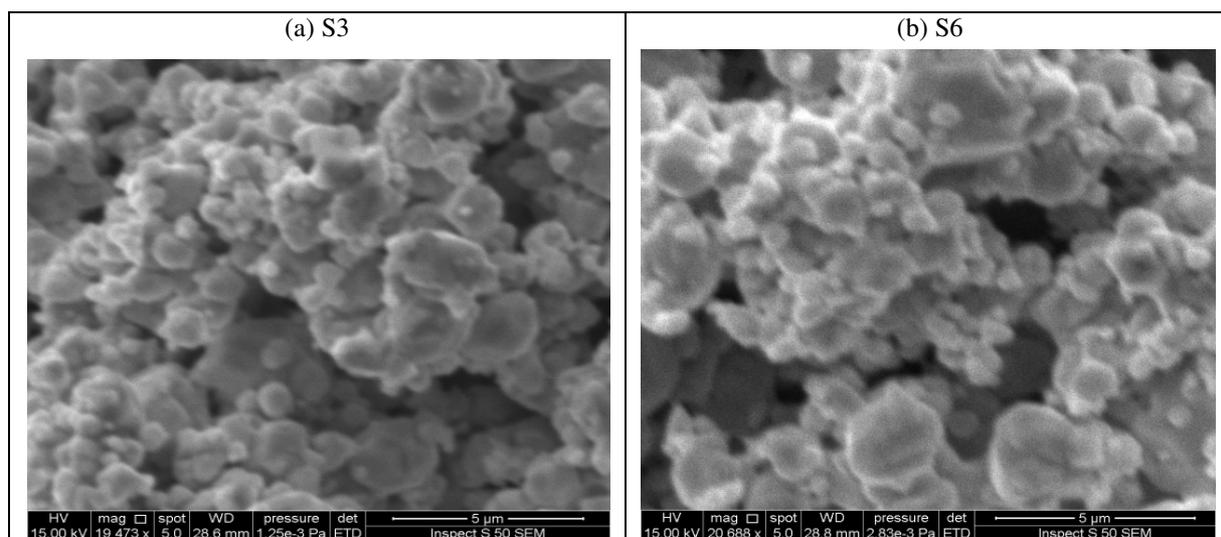


Figure 7. SEM images of Cr_2O_3 -NPs

4- Conclusions

Chromium oxide nanoparticles (Cr_2O_3 -NPs) with a rhombohedral structure have been prepared successfully by sol-gel method using $\text{Cr}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$ as precursor material, triethanolamine (TEA) as template and water as green solvent under microwave irradiation. The results show that the average crystallite size are estimated by D-S formula and W-H formula of (Cr_2O_3 -NPs) for all samples in nanoscale. The average crystallite size increase with the increase of annealing temperature and concentration of (TEA). The crystal lattice distortion ratio $\langle \varepsilon^2 \rangle^{1/2}$ and the specific surface area (SSA) are inversely proportional, dislocation density (δ) and the number of unit cells (n) are indirectly proportional to particle size. The obtained results have a good optimization between the observed X-ray diffraction patterns and that are calculated by Rietveld analysis. FTIR spectra validated the purity of Cr_2O_3 -NPs. The SEM micrographs of Cr_2O_3 -NPs for S3 and S6 show all particles exhibit a spherical shape with a high degree of agglomeration among fine particles.

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